

## Electronic Supplementary Information

# Enhancement of Oxygen Surface Exchange on Epitaxial $\text{La}_{0.6}\text{Sr}_{0.4}\text{Co}_{0.2}\text{Fe}_{0.8}\text{O}_{3-\delta}$ Thin Films using Advanced Heterostructured Oxide Interface Engineering

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## Experimental details

**Target Synthesis.** Both  $\text{La}_{0.6}\text{Sr}_{0.4}\text{Co}_{0.2}\text{Fe}_{0.8}\text{O}_{3-\delta}$  ( $\text{LSCF}_{113}$ ) and  $\text{LaSrCoO}_{4\pm\delta}$  ( $\text{LSC}_{214}$ ) were prepared by the Pechini methods.  $\text{La}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ ,  $\text{Co}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ ,  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ ,  $\text{Sr}(\text{NO}_3)_2$ , and separately  $\text{La}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ ,  $\text{Co}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ ,  $\text{Sr}(\text{NO}_3)_2$  were dissolved in de-ionized water with ethylene glycol, and citric acid (Sigma-Aldrich, USA) mixture to synthesize  $\text{LSCF}_{113}$  and  $\text{LSC}_{214}$  respectively. After esterification at 100 °C, the resin was charred at 400 °C and finally calcined at 1000 °C in air for 12 hours. The  $\text{La}_{0.8}\text{Sr}_{0.2}\text{CoO}_{3-\delta}$  ( $\text{LSC}_{113}$ ) and  $\text{Gd}_{0.2}\text{Ce}_{0.8}\text{O}_2$  (GDC) were also prepared by the Pechini methods<sup>1</sup>.  $\text{La}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ ,  $\text{Sr}(\text{NO}_3)_2$ ,  $\text{Mn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ , and separately  $\text{Gd}(\text{NO}_3)_3$  and  $\text{Ce}(\text{NO}_3)_3$  were dissolved in de-ionized water with ethylene glycol, and citric acid (Sigma-Aldrich, USA) mixture to synthesize  $\text{LSC}_{113}$  and GDC respectively. After esterification at 100 °C, the resin was charred at 400 °C and finally calcined at 1000 °C in air for 12 hours. Pulsed laser deposition (PLD) target pellets with 25 mm diameter were subsequently fabricated by uniaxial pressing at 50 MPa. The  $\text{LSCF}_{113}$ ,  $\text{LSC}_{113}$ ,  $\text{LSC}_{214}$ , and GDC pellets were fully sintered at 1,300 °C in air for 6 hours, 1,200 °C in air for 10 hours, 1,350 °C in air for 12 hours, and 1,100 °C in air for 14 hours, respectively.

**Sample preparation.** Single crystal 9.5 mol%  $\text{Y}_2\text{O}_3$ -stabilized  $\text{ZrO}_2$  (YSZ) wafers with (001) orientation and dimensions of  $10 \times 5 \times 0.5$  mm (MTI corporation, USA), were used as substrate. Prior to  $\text{LSC}_{214}$ ,  $\text{LSCF}_{113}$ ,  $\text{LSC}_{113}$ , and GDC deposition, platinum ink (Pt) (#6082, BASF, USA) counter electrodes were painted on one side of the YSZ and dried at 900 °C in air for 1 hour. PLD was performed using a KrF excimer laser at  $\lambda = 248$  nm, 10 Hz pulse rate and 45 mJ pulse energy under  $p(\text{O}_2)$  of 50mTorr with 500 pulses of GDC (~5 nm) at 550 °C, followed by 15,000

pulses of LSCF<sub>113</sub> (~63 nm) at 650 °C. Immediately after completing the LSCF<sub>113</sub> base film deposition, LSC<sub>214</sub> and LSC<sub>113</sub> films were subsequently deposited. After completing the final deposition, the sample was cooled to room temperature in the PLD chamber for ~1 hour under an oxygen partial pressure of 50 mTorr. The film thicknesses were determined by atomic force microscopy (AFM). The utilization of reflection high-energy electron diffraction (RHEED) enabled diagnostic *in-situ* monitoring of the LSCF<sub>113</sub> film growth.

**Relaxed lattice parameter determination by HRXRD.** The Relaxed lattice parameter  $\hat{a}$  and  $\hat{c}$  are derived from the following equation (where  $\hat{a}$  and  $\hat{c}$  are the relaxed lattice parameters for the film in an unstrained state),<sup>2-4</sup>  $\frac{(c-\hat{c})}{\hat{c}} = \frac{-2\nu (a-\hat{a})}{1-\nu \hat{a}}$ , assuming  $\hat{a} = \hat{c}$ , and  $\nu = 0.25$ .<sup>2</sup> The in-plane strain is given by:  $\epsilon_{aa} = \frac{(a-\hat{a})}{\hat{a}}$  and the out of plane strain by:  $\epsilon_{cc} = \frac{(c-\hat{c})}{\hat{c}}$

**Microelectrodes Fabrication.** *In situ* electrochemical impedance spectroscopy (EIS) measurements were conducted to probe ORR activity on geometrically well-defined LSCF<sub>113</sub> with single-layer and double-layer decoration microelectrodes fabricated by photolithography and acid etching, where sintered porous Pt sintered onto the backside of the YSZ substrate served as the counter electrode. OCG positive photoresist (Arch Chemical Co., USA) was applied on the surface of the samples and patterned using a mask aligner (Karl Süss, Germany,  $\lambda = 365$  nm). The photoresist was developed using Developer 934 1:1 (Arch Chemical Co., USA) and the thin films were etched in hydrochloric acid (HCl) to remove the LSCF<sub>113</sub> with single-layer and double layer decoration films excess and create the circular microelectrodes (diameters ~50  $\mu\text{m}$ , ~100  $\mu\text{m}$ , ~150  $\mu\text{m}$ , and ~200  $\mu\text{m}$ , exact diameter determined by optical microscopy). Before

electrochemical testing, microelectrode geometry and morphology was examined by optical microscopy (Carl Zeiss, Germany) and atomic force microscopy (AFM) (Veeco, USA).

**Electrochemical Characterization.** Fig. S4(b) and S4(c) detail the equivalent circuit and corresponding Nyquist plot for this experimental system. *ZView* software (Scribner Associates, USA) was used to construct the equivalent circuit and perform complex least squares fitting. The EIS data were fitted using a standard resistor ( $R_1$ ) for *HF* and resistors ( $R_2$ ) in parallel with a constant phase elements ( $CPE_2$ ) for *MF* and *LF* ( $R_1-(R_2/CPE_2)-(R_{ORR}/CPE_{ORR})$ ). Based on the  $p(O_2)$  dependence of the three features, physical or chemical process with regard to each frequency range can be determined.<sup>5-8</sup> The *HF* feature was found unchanged with  $p(O_2)$ , and its magnitude and activation energy ( $\sim 1.15$  eV) were comparable to those of oxygen ion conduction in YSZ reported previously<sup>9</sup>. The *MF* feature, which was found to have a  $p(O_2)$  independent feature, was attributed to interfacial transport of oxygen ions between the LSCF<sub>113</sub> film and the GDC layer. In addition, the magnitude of its capacitance was relatively small ( $\sim 10^{-6}$  F) compared to the *LF* feature ( $\sim 10^{-3}$  F). The *LF* feature was found to have a strong  $p(O_2)$  dependence. The resistance of the *LF* feature drastically increases as oxygen partial pressure decreases. In the case of thin film samples, the magnitude of capacitance is due to the oxygen content change in the films. Therefore, the electrode oxygen surface reaction corresponds with the *LF* feature. We obtained values for  $R_{ORR}$  and knowing the area of the microelectrode ( $A_{\text{electrode}} = 0.25 \pi d_{\text{electrode}}^2$ ). Then, we can determine the ORR area specific resistance ( $ASR_{ORR} = R_{ORR} \cdot A_{\text{electrode}}$ ). The electrical surface exchange coefficient ( $k^q$ ), which is comparable to  $k^*$ ,<sup>10</sup> was determined using the expression,<sup>11, 12</sup>

$$k^q = RT / 4F^2 R_{ORR} A_{\text{electrode}} c_O \quad (1)$$

where  $R$  is the universal gas constant ( $8.314 \text{ J mol}^{-1} \text{ K}^{-1}$ ),  $T$  is the absolute temperature,  $F$  is the Faraday's constant ( $96,500 \text{ C mol}^{-1}$ ), and  $c_0$  is the lattice oxygen concentration in  $\text{LSCF}_{113}$  where

$$c_0 = (3-\delta)/V_m, \quad (2)$$

$V_m$  is the molar volume of  $\text{LSCF}_{113}$  at room temperature. In this study,  $c_0$  was calculated with  $\delta$  extracted from previous reported values<sup>13</sup>.

The surface coverage may change the  $c_0$  value of the system. For estimating this influence, we compared  $\text{LSCF}_{113}$   $c_0$  values with  $\text{LSC}_{113}$   $c_0$  values. However, calculated  $c_0$  values for  $\text{LSCF}_{113}$  was only  $\sim 1 - 2 \%$  different from those for  $\text{LSC}_{113}$ . We therefore decide to use  $c_0$  values for  $\text{LSCF}_{113}$  for all samples.

$VSC$ , indicative of changes in the oxygen nonstoichiometry induced by changes in the electrical potential, can be obtained from EIS data via the expression<sup>14</sup>

$$VSC = [1/(A_{\text{electrode}} \cdot \text{thickness})][(R_{\text{ORR}})^{1-n}Q]^{1/n}, \quad (3)$$

where  $Q$  is the non-ideal "capacitance", and  $n$  is the non-ideality factor of  $CPE$ . The fitted values of  $n$  for semi-circle  $CPE_{\text{ORR}}$  were found to range from  $\sim 0.96$  to  $1.0$  over the entire  $p\text{O}_2$  range examined ( $n=1$ , ideal).

Table S1. Constrained and relaxed lattice parameters of LSCF<sub>113</sub>, LSCF<sub>113</sub> with the ~3 nm single-layer decoration of mixed LSC<sub>214</sub> and LSC<sub>113</sub> and the double-layer decoration of stacked ~3 nm LSC<sub>214</sub> and ~0.5 nm LSC<sub>113</sub> extracted from normal and off-normal XRD data at room temperature. Constrained normal and in-plane lattice parameters of all films were calculated from combining the inter-planar distance of the LSCF<sub>113</sub> (002)<sub>pc</sub> and LSCF<sub>113</sub> (202)<sub>pc</sub> peaks.

Samples	Constrained in-plane $a / \text{\AA}$	Constrained normal $c / \text{\AA}$	Relaxed lattice parameter $\hat{a} / \text{\AA}$	In-plane strain $\epsilon_{aa} = \frac{(a - \hat{a})}{\hat{a}} / \%$	Normal strain $\epsilon_{cc} = \frac{(c - \hat{a})}{\hat{a}} / \%$
LSCF ref.	3.897	3.899	3.898	-0.038	0.025
LSC <sub>214</sub>	3.897	3.902	3.900	-0.077	0.051
LSC82 <sub>113</sub> :LSC <sub>214</sub>	3.896	3.888	3.891	-0.013	0.088
LSC64 <sub>113</sub> :LSC <sub>214</sub>	3.893	3.897	3.896	-0.068	0.045
LSC46 <sub>113</sub> :LSC <sub>214</sub>	3.891	3.906	3.900	-0.023	0.015
LSC82 <sub>113</sub> + LSC <sub>214</sub>	3.905	3.903	3.903	-0.028	0.019
LSC64 <sub>113</sub> + LSC <sub>214</sub>	3.893	3.912	3.904	-0.029	0.019
LSC46 <sub>113</sub> + LSC <sub>214</sub>	3.899	3.907	3.904	-0.013	0.089

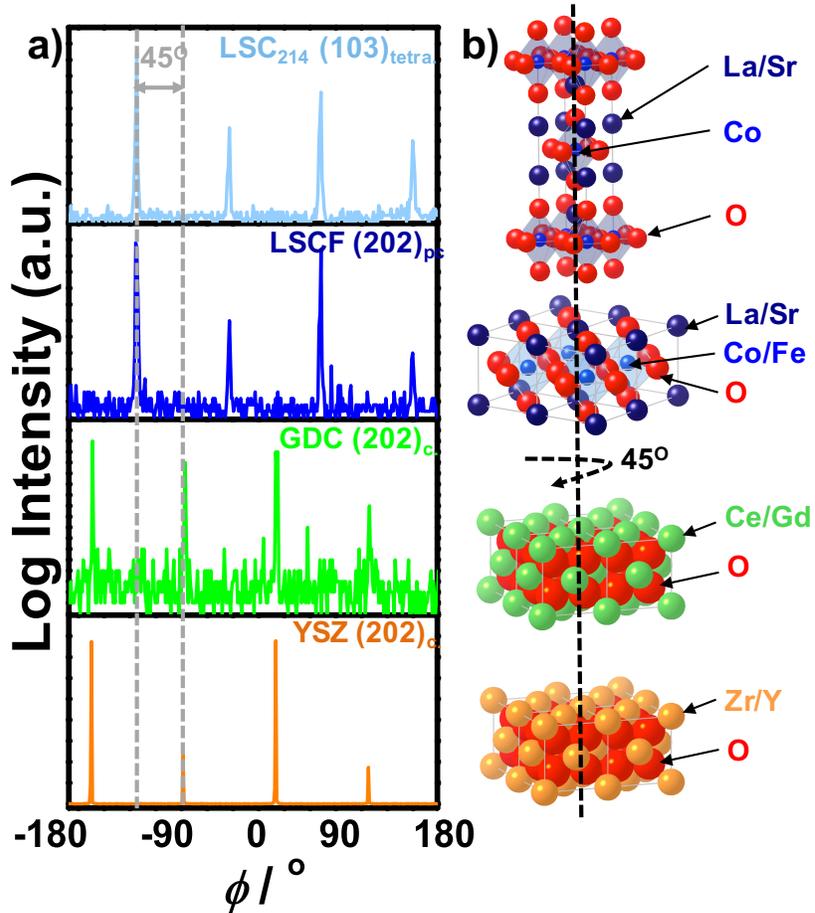


Figure S1. (a) off-normal XRD of a  $\text{LSCF}_{113}$  sample with  $\sim 3$  nm  $\text{LSC}_{214}$  coverage, and (b) schematic of the crystallographic rotational relationships among the  $\text{LSC}_{214}(001)_{\text{tetra}}$ ,  $\text{LSCF}_{113}(001)_{\text{pc}}$ ,  $\text{GDC}(001)_{\text{cubic}}$ , and  $\text{YSZ}(001)_{\text{cubic}}$ .

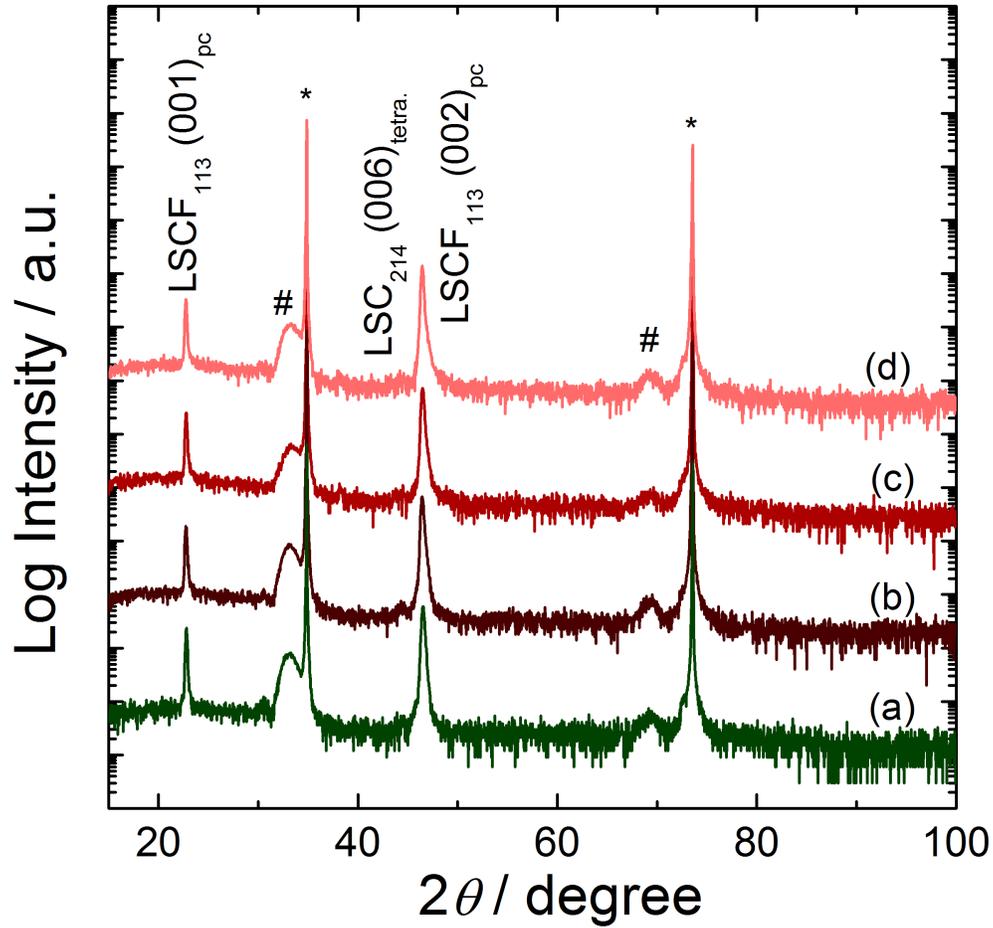


Figure S2. High-resolution X-ray diffraction analysis of (a) the  $\text{LSCF}_{113}$  reference (green), (b) the  $\text{LSCF}_{113}$  with double-layer decorations of stacked  $\text{LSC}_{46_{113}} 25\%:\text{LSC}_{214} 75\%$  (dark red), (c)  $\text{LSC}_{46_{113}} 50\%:\text{LSC}_{214} 50\%$  (red), and  $\text{LSC}_{46_{113}} 75\%:\text{LSC}_{214} 25\%$  (light red) epitaxial thin films on (001) YSZ substrates with GDC buffer layer. YSZ substrate and GDC peaks are indicated with pounds (#) and asterisks (\*), respectively.

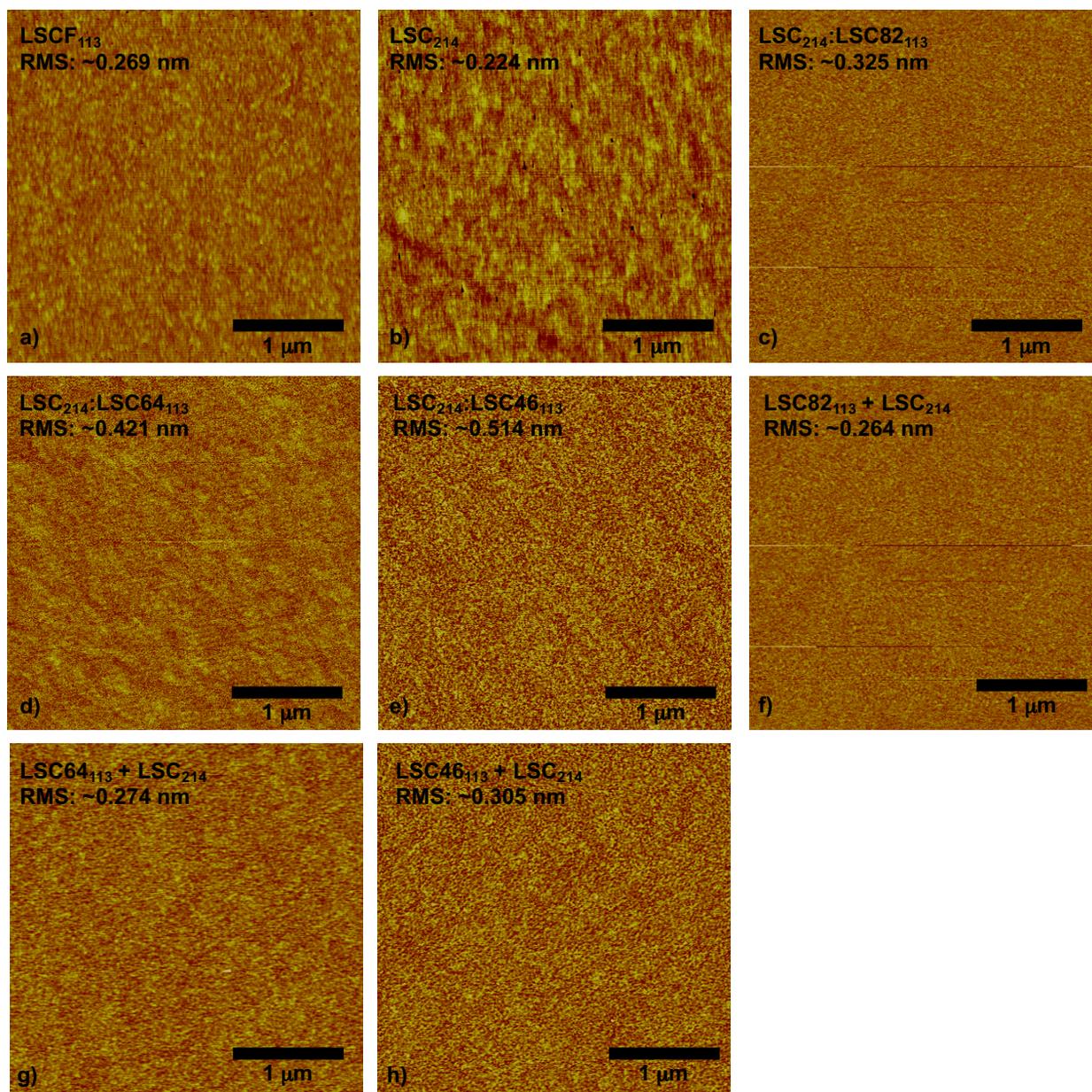


Figure S3. AFM images of (a) the  $\sim 65$  nm  $\text{LSCF}_{113}$  reference, (b) the  $\sim 3$  nm  $\text{LSC}_{214}$ -decorated  $\text{LSCF}_{113}$ , and (c) the  $\text{LSCF}_{113}$  with  $\sim 3$  nm single layer decorations of mixed  $\text{LSC}_{214}$  and  $\text{LSC}_{82113}$ , (d)  $\text{LSC}_{64113}$ , and (e)  $\text{LSC}_{46113}$  and the  $\text{LSCF}_{113}$  with double-layer decorations of stacked  $\sim 3$  nm  $\text{LSC}_{214}$  and (f)  $\sim 0.5$  nm  $\text{LSC}_{82113}$ , (g)  $\sim 0.5$  nm  $\text{LSC}_{64113}$ , and (h)  $\sim 0.5$  nm  $\text{LSC}_{46113}$  epitaxial thin films on (001) YSZ substrates with GDC buffer layer.

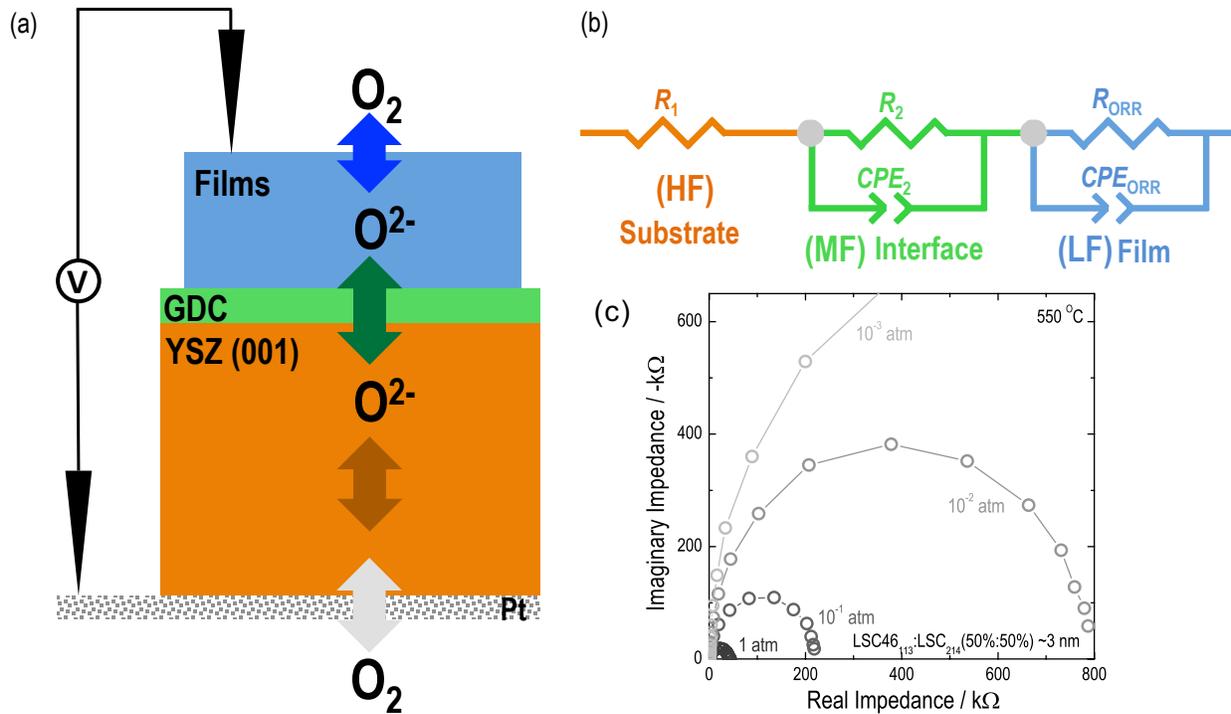


Figure S4. (a) Schematic of a LSCF/GDC/YSZ(001)/porous Pt samples and electrochemical testing configuration (not drawn to scale), and (b) equivalent circuit ( $R_1$  = YSZ electrolyte resistance,  $R_2$  = electrode/electrolyte interface resistance<sup>4</sup>,  $R_{\text{ORR}}$  = ORR resistance, CPE = constant phase element) used to extract ORR kinetics, and (c) Nyquist plot of the LSCF<sub>113</sub> with a single layer decoration of LSC<sub>214</sub> mixed with LSC<sub>46113</sub> thin films at 550 °C;

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